

by Gschneidner in terms of a considerable homogeneity region. In order to check this interpretation the X-ray powder diagrams of three arc-cast samples annealed at 800°C and containing slightly different relative amounts of La and Al were compared (Table 4). The first (20.0 at.% La) was Buschow's original diagram, which had served for his determination of the ' $\alpha$ -LaAl<sub>4</sub>' lattice constants; besides the La<sub>3</sub>Al<sub>11</sub> lines, there are also weak lines of Al. The second, corresponding to the exact stoichiometric composition (21.4 at.% La) shows no other lines than those of pure La<sub>3</sub>Al<sub>11</sub>, whereas the third (22.2 at.% La) also contains LaAl<sub>3</sub> lines. This proves not only that the homogeneity region is rather small but also that La<sub>3</sub>Al<sub>11</sub> is indeed the correct chemical formula. The invariance of the lattice constants tends to corroborate the former conclusion.

Since the high-temperature modification cannot be quenched (Buschow), one would think that all previous investigations, which were carried out at room temperature, would refer to the low-temperature phase. The question then arises why the orthorhombic structure had not earlier been recognized. Poor resolution of the X-ray diagrams might be held responsible, but such an explanation seems particularly unsatisfactory in the case of Rossi's work, which involves tetragonal Laue photographs of single-crystalline samples. We are inclined to believe, therefore, that the BaAl<sub>4</sub>-type structure is stabilized by chemical impurities. Although this idea needs experimental verification, it should be noted that purer starting materials were available to us (La 99.9%, Al 99.99%) than to any of the previous investigators, and also that our method of sample prep-

aration was probably less apt to introduce contamination.

Since it is impossible to quench the high-temperature modification, the  $\alpha$  and the  $\beta$  phases are likely to have the same chemical composition, in which case they should be designated as  $\alpha$ -La<sub>3</sub>Al<sub>11</sub> and  $\beta$ -La<sub>3</sub>Al<sub>11</sub>. If so, and if Buschow's statement is correct that ' $\beta$ -LaAl<sub>4</sub>' has the BaAl<sub>4</sub>-type structure, the phase transition entails the ordering of vacancies by diffusion. This conclusion is not necessarily at variance with the reported sharpness (Buschow, 1965a) of the transition.

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### Crystal and Molecular Structure of *S,S*-Dimethyl-*N*-methylsulphonylsulphilimine, (CH<sub>3</sub>)<sub>2</sub>SNSO<sub>2</sub>CH<sub>3</sub>

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The crystal structure of *S,S*-dimethyl-*N*-methylsulphonylsulphilimine has been solved by three-dimensional Patterson and Fourier calculations. The parameters have been refined by the aid of Booth's differential syntheses, with a final  $R=0.106$ . The S-N distances [1.58<sub>1</sub>(10) and 1.63<sub>3</sub>(9) Å] indicate a delocalized S-N-S  $d_{\pi}$  bond system. The S-N-S bond angle is 116.2 (6)°, the average S-O distance, 1.44<sub>6</sub> Å. The C(1)-S(VI) bond distance, 1.74<sub>9</sub> (12) is similar to other C(sp<sup>3</sup>)-S(VI) bonds. The C(sp<sup>3</sup>)-S(IV) bond distances [1.74 (7) and 1.74 (3) Å] are rather short, which can be explained only with the supposition of strong hyperconjugation. The large e.s.d.'s of the C(2)-S(IV) and C(3)-S(IV) bonds are due to the intensive thermal motion of the C(2) and C(3) atoms.

#### Introduction

The structure and the stereochemistry of the *N*-acrylsulphilimines have been investigated by Kucsman and his collaborators (Kucsman, 1953, 1958; Kucsman &

Kapovits, 1964). These investigations gave interesting results concerning the bond system and the configuration of these compounds. Among others the presence of a strong S(IV)-N  $d_{\pi}$  bond in the *N*-sulphonylsulphilimines (RR'SNSO<sub>2</sub>Q) has been proved by infrared

spectroscopy (Kuczman, Ruff & Kapovits, 1966). It has not, however, been possible to draw any unambiguous conclusion concerning the character of the S(VI)–N bond from spectroscopic data. Moreover, in the infrared spectrum of the sulphilimines containing an S(IV)CH<sub>3</sub> group an anomalous phenomenon was revealed, which was described as the ‘*S*-methyl effect’.

To clear up this problem the complete structure determination of one of the sulphilimines (R,R' and Q=CH<sub>3</sub>) has been performed, as a first step towards the systematic structure analysis of the sulphilimine derivatives.

### Experimental

The crystals of the compound were prepared by the method of Schulz & Kresze (1963) and were recrystallized from a mixture of absolute benzene and alcohol. According to the infrared spectrum the colourless needle-like crystals are free from mother liquor impurities. The m.p. (122°C) is in good agreement with the literature data.

#### Crystal data

C<sub>3</sub>H<sub>9</sub>NO<sub>2</sub>S<sub>2</sub>, *M* = 155.25

*a* = 5.805 ± 0.012, *b* = 13.503 ± 0.009, *c* = 9.213 ± 0.012 Å

*β* = 105.15 ± 0.13°, *V* = 697.07 Å<sup>3</sup>, *Z* = 4,

*D<sub>x</sub>* = 1.479 g.cm<sup>-3</sup>

*D<sub>m</sub>* = 1.473 g.cm<sup>-3</sup> (by flotation).

*μ* = 60 cm<sup>-1</sup> (Cu *Kα*), *F*<sub>000</sub> = 328,

Space group *P*2<sub>1</sub>/*c* (from systematic absences).

The lattice parameters were determined by the precession camera, calibrated with sodium chloride. The intensities were measured by a Zeiss fast photometer on integrated, equi-inclination Weissenberg photographs, taken with Ni-filtered Cu *Kα* radiation, using the multiple-film technique. The dimensions of the crystals investigated were reduced by our preparation device (Kálmán & Argay, 1965) below the value of *μR* = 0.4, and in this way the absorption correction could be neglected. Unfortunately the crystals slowly decomposed during the long X-ray irradiation. Therefore the investigated samples were changed after every second layer. It was hoped that the probable effect of

this phenomenon upon the values of the intensities would be diminished by collection of data from repeated photographs, made around the three crystal axes, and controlling them by each other.

Photographs were taken from the following layers:

*hk*0, . . . . . *hk*6,  
*h*0*l*, . . . . . *h*4*l*,  
*Ok**l*, *1kl*\* and *2kl*\*

The spot shape correction on the non-equatorial layers and the multiple-film scaling were taken by hand calculations, measuring the intensities on both upper and lower parts of the films. Then the intensities were corrected for the *Lp* factor by a National Elliott 803B computer, using the program of Sasvári & Sánta (1964). After the interlayer scaling, made by hand, the absolute scale and the approximate temperature factors (*B̄* = 3.63 Å<sup>2</sup>) were determined by Wilson's method. In this way 1264 independent reflexions (81% of the possible ones) were collected. For the structure determination and refinement the 1027 non-zero reflexions were used.

### Structure analysis and refinement

The interpretation of the 3-D Patterson calculation using the ‘satellite and rotation vector’s relations’ (Buerger, 1959) gave unambiguously all sulphur–sulphur vectors. The (*u, v, w*) Patterson coordinates were transformed to the (*x, y, z*) Fourier ones by the aid of the shortest vector, the absolute value of which was in good agreement with the estimated intramolecular S . . . S distance. The first 3-D Fourier synthesis was performed with the phases appropriate to the sulphur atoms. Three successive rounds of structure factor and Fourier calculations gave the positional parameters of all the atoms; *R* was now 0.231.

The refinement was carried out with twelve cycles of Booth's differential synthesis (1948), two with the average, two with atomic isotropic and eight with anisotropic thermal parameters. The latter were calculated by the method of Nardelli & Fava (1960), using the

\* These intensities were used only to control the same ones collected from other layers.

Table 1. Final positional (monoclinic\*) parameters with their standard deviations

	<i>x</i>	<i>y</i>	<i>z</i>	<i>σ</i> ( <i>x</i> )	<i>σ</i> ( <i>y</i> )	<i>σ</i> ( <i>z</i> )
S(1)	0.1301	0.1175	0.2137	0.0004	0.0001	0.0003
S(2)	0.1640	−0.0785	0.2886	0.0005	0.0001	0.0003
O(1)	0.1705	0.2100	0.2936	0.0021	0.0006	0.0012
O(2)	−0.1150	0.0963	0.1313	0.0017	0.0007	0.0013
N	0.2509	0.0353	0.3312	0.0017	0.0006	0.0012
C(1)	0.2913	0.1206	0.0777	0.0020	0.0006	0.0012
C(2)	0.4055	−0.1399	0.2515	0.0144!	0.0019	0.0068!
C(3)	0.1800	−0.1305	0.4639	0.0039!	0.0009	0.0039!

\* For the calculation of the clinographical projections (Figs. 1 and 2) the orthogonalized parameters were used. The transformation matrix from monoclinic to orthogonal coordinates was:

$$\begin{pmatrix} 1 & 0 & \cos \beta \\ 0 & 1 & 0 \\ 0 & 0 & \sin \beta \end{pmatrix}$$

second derivatives of the electron densities from the differential syntheses. In view of the uncertainty in deriving the correct scale factors, the anisotropic thermal parameters, given in Table 2 have only limited physical meaning.

Because of the anomalous differences in  $\rho$  values of the carbon atoms, a 3-D ( $F_o-F_c$ ) synthesis was performed to verify the positional parameters of the C(2) and C(3) atoms. The result of the difference synthesis confirmed the coordinates of these atoms, obtained from the third 3-D Fourier synthesis. To find the reason of the difference in  $\rho$  values of the carbon atoms, the ( $x,y$ ) and ( $x,z$ ) projections of their electron densities were drawn, on the basis of the third 3-D Fourier cycle. These projections using an arbitrary limit of the electron density ( $\rho_0 \geq 3 \text{ e.}\text{\AA}^{-3}$ ) show that while C(1) has an almost spherical electron cloud, C(3) and especially C(2) have unusually deformed elliptical ones. The estimated volumes of these electron clouds are inversely proportional to the electron densities found in the atomic centres. These results are consistent with each other and with the unusually great thermal parameters of the C(2) and C(3) atoms.

After the seventh cycle of the refinement a correction of the accidental errors and of the effect of the secondary extinction upon the intensities was performed. Twenty reflexions in the centre of the reciprocal lattice ( $\sin^2\theta/\lambda^2 \leq 0.03$ ) were corrected by the method of Pinnock, Taylor & Lipson (1956).

The final coordinates with their e.s.d.'s (Cruickshank, 1949) are given in Table 1. In Table 3 are listed the observed and calculated values of the electron density and the second derivatives at the atomic peaks with e.s.d.'s. The determination of the hydrogen coordinates was not attempted, because of the great thermal motion of the methyl groups. The final reliability indices,  $R$  (for the observed reflexions only) and  $R'$  (including the

Table 2. Anisotropic thermal parameters ( $\text{\AA}^2$ )

The parameters are values of  $B_{ij}$  in the expression:

$$\exp[-\frac{1}{4}(B_{11}a^2h^2 + B_{22}b^2k^2 + B_{33}c^2l^2 + 2B_{23}b^*c^*kl + 2B_{13}a^*c^*hl + 2B_{12}a^*b^*hk)]$$

	$B_{11}$	$B_{22}$	$B_{33}$	$B_{23}$	$B_{13}$	$B_{12}$
S(1)	2.93	2.47	2.93	0.01	1.28	0.12
S(2)	4.64	2.80	2.96	-0.18	0.42	0.33
O(1)	8.89	2.75	4.64	-0.08	3.38	-0.22
O(2)	2.88	4.40	6.90	1.70	1.22	0.60
N	4.15	3.13	3.21	0.20	0.89	-0.23
C(1)	3.96	4.34	3.76	0.08	1.75	-0.33
C(2)	16.20	6.25	9.67	1.05	9.05	4.17
C(3)	6.98	3.41	4.25	-0.02	1.77	-0.32

non-observed ones, using  $F_o = 0.5F_{\text{min}}$ , when  $F_c \geq F_{\text{min}}$  are 0.106 and 0.128, respectively.

The scattering factors used throughout the calculations were those of Berghuis, Haanappel, Potters, Loopstra, MacGillavry & Veenendaal (1955) for oxygen, nitrogen and carbon, and of Dawson (1960) for sulphur. The calculations were performed on the Olivetti Elea 6001/S computer of the Centro di Calcolo Elettronico of the University of Parma, using the programs of Nardelli, Musatti, Domiano & Andreotti (1964, 1965).

### Discussion

The atomic distances and angles are given in Table 4. The e.s.d.'s are calculated by the formula of Ahmed & Cruickshank (1953) for distances and from those of Darlow (1960) for angles. The clinographic projections of the molecule are shown in Figs. 1 and 2.

The sulphur atoms with different valence states [S(IV) and S(VI)] are linked by an  $sp^2$ -nitrogen atom, which is able to establish  $\pi$ -connexions. The S-N distances (1.58<sub>1</sub> and 1.63<sub>3</sub>  $\text{\AA}$ ) indicate that a delocalized  $d_\pi$  bond system is formed on the S-N-S atoms with strong  $d_\pi$  bonds. The deviation between them is signi-

Table 3. Comparison of peak heights ( $\text{e.}\text{\AA}^{-3}$ ) and curvatures ( $\text{e.}\text{\AA}^{-5}$ ) from the differential synthesis

		$\rho$	$-A_{hh}$	$-A_{kk}$	$-A_{ll}$	$A_{hk}$	$A_{hl}$	$A_{kl}$
S(1)	obs.	31.0	309	308	245	4	79	-15
	calc.	30.9	308	308	245	4	78	-14
S(2)	obs.	26.7	226	264	209	11	40	-18
	calc.	26.7	227	264	208	11	40	-18
O(1)	obs.	8.8	55	70	57	-2	12	18
	calc.	8.8	57	70	57	-3	12	19
O(2)	obs.	8.7	83	58	60	-2	25	1
	calc.	8.7	82	58	60	-2	25	1
N	obs.	8.5	67	60	56	-3	13	0
	calc.	8.6	67	60	56	-3	13	0
C(1)	obs.	7.4	61	66	59	1	16	-2
	calc.	7.3	61	67	59	1	16	-1
C(2)	obs.	4.0	16	34	19	8	10	-5
	calc.	3.9	16	33	19	8	10	-6
C(3)	obs.	5.7	39	46	18	-7	8	1
	calc.	5.6	39	46	19	-8	8	2
	e.s.d.	0.2	2	3	2	1	1	1

ficant according to the significance test of Cruickshank & Robertson (1953):

$$t_0 = (l_1 - l_2)(\sigma_1^2 + \sigma_2^2)^{-1/2} = 3.54,$$

but can be explained on the basis of the different atomic radii and electronegativities of the S(IV) and S(VI) atoms (Truter, 1962b).

The S–O distances [1.43<sub>8</sub> and 1.45<sub>5</sub> Å] indicate strong bonds comparable to those found in other sulphones and sulphonamides. The C(1)–S(VI) bond, 1.74<sub>9</sub> Å, is not significantly shorter ( $t_{0, \max} = 1.64$ ) than other C(sp<sup>3</sup>)–S(VI) bonds, e.g.

1.770 ± 0.009 Å in (CH<sub>3</sub>SO<sub>2</sub>)<sub>2</sub>C=C=NCH<sub>3</sub>  
(Wheatley, 1954)

1.761 ± 0.01 (CH<sub>3</sub>SO<sub>2</sub>)(C<sub>6</sub>H<sub>5</sub>SO<sub>2</sub>)C=C=NCH<sub>3</sub>  
(Bullough & Wheatley, 1957)

1.770 ± 0.007 K<sub>2</sub>CH<sub>2</sub>(SO<sub>3</sub>)<sub>2</sub>  
(Truter, 1962a).

The C(2)–S(IV) and C(3)–S(IV) bonds are considerably shorter than the theoretically calculated C(sp<sup>3</sup>)–S(IV) or the one found, for example, in rongalite, 1.83 ± 0.01 Å (Truter, 1962b). These unusually short distances, though their standard deviations are rather high, seem to originate from a hyperconjugation-like effect. This question nevertheless needs the thermal motion analysis of atoms as well as of the molecule, and this could not be performed at present because of lack of computer facilities. At any rate, the 'S-methyl effect' mentioned above should clearly be associated with the shortening of the C(sp<sup>3</sup>)–S(IV) bond lengths and the intensive thermal motion of the methyl groups. The thermal vibration of the C(2) is considerably greater than that of C(3), for which, however, an unambiguous explanation could not be found.

Another remarkable fact is that the standard deviations of the S–O bond lengths are not influenced as significantly by the great thermal motion of the oxygen atoms as that of the C–S(IV) bond lengths by that of the carbon atoms.

The S(IV)–N–S(VI) bond angle, 116.2°, is less than the theoretical 120°. The lone electron pair of the nitrogen, which does not take part in the S–N–S delocalized *d<sub>π</sub>* system, presumably displays a repulsion effect upon the S–N bonds, and therefore the bond angle decreases. The O(1)–S(VI)–O(2), O(1)–S(VI)–C(1) and O(2)–S(VI)–C(1) bond angles are similar to the corresponding ones in the CH<sub>3</sub>SO<sub>2</sub> groups of other

compounds investigated (Wheatley, 1954; Bullough & Wheatley, 1957). The O(2)–S(VI)–N angle is considerably greater than the O(1)–S(VI)–N angle and those found in potassium imidodisulphate (Cruickshank & Jones, 1963) in the dinitrosulphite and sulphamate ions (Jeffrey & Stadler, 1951). The bond angles of the

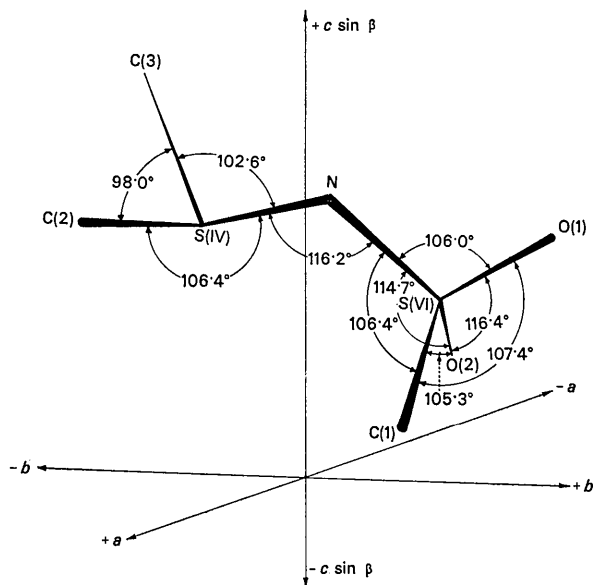


Fig. 1. The atomic distances and labelling of the atoms in sulphilimine.

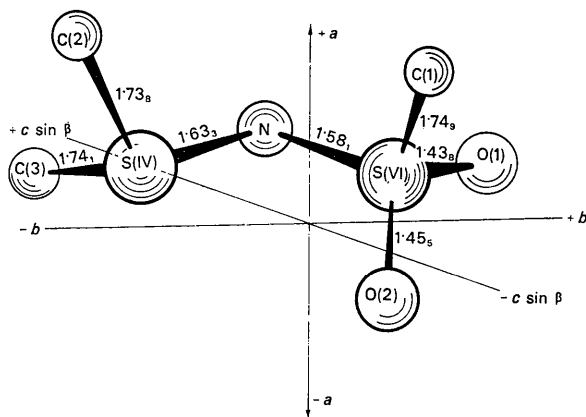


Fig. 2. The bond angles and labelling of the atoms in sulphilimine.

Table 4. Interatomic distances and bond angles with their e.s.d.'s

S(VI)–O(1)	1.438 ± 0.009 Å	S(VI)–N–S(IV)	116.2 ± 0.6°
S(VI)–O(2)	1.455 ± 0.012	O(1)–S(VI)–O(2)	116.4 ± 0.6
S(VI)–N	1.581 ± 0.010	O(1)–S(VI)–N	106.0 ± 0.6
S(VI)–C(1)	1.749 ± 0.012	O(1)–S(VI)–C(1)	107.4 ± 0.6
S(IV)–N	1.633 ± 0.009	O(2)–S(VI)–N	114.7 ± 0.6
S(IV)–C(2)	1.74 ± 0.07	O(2)–S(VI)–C(1)	105.3 ± 0.6
S(IV)–C(3)	1.74 ± 0.03	N–S(VI)–C(1)	106.4 ± 0.5
S···S	2.729 ± 0.003	N–S(IV)–C(2)	106.4 ± 1.7
		N–S(IV)–C(3)	102.6 ± 0.7
		C(2)–S(IV)–C(3)	98.0 ± 1.9

CH<sub>3</sub>S(IV)-N group can be compared with those of dimethyl sulphoxide (Thomas, Shoemaker & Eriks, 1966) which has a similar electron structure:

	Sulphil-		Sulph-
	imine		oxide
C(2)-S(IV)-C(3)	98.0°	C(1)-S(IV)-C(2)	98.2°
C(2)-S(IV)-N	106.4	C(1)-S(IV)-O	106.7
C(3)-S(IV)-N	102.6	C(2)-S(IV)-O	107.5

As has been emphasized throughout this paper, some atoms of the molecule had unusually great thermal vibrations. This question suggests that the structure should be reinvestigated and refined with more accurate intensities collected at low temperature, as suggested by Cruickshank (1960) and Jensen (1962) in such

cases. The observed slow decomposition of the crystals during X-ray irradiation may also be connected with the great thermal motion of the structure. Perhaps an electron spin resonance study of the irradiated crystals will give some information concerning this phenomenon.

The conformation of the molecule from the organic chemical aspect will be discussed elsewhere (Kuczman, Kálmán & Kapovits, 1966).

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Table 5. Observed and calculated structure factors

Unobserved reflexions are marked by a dot.

h	k	l	IOF <sub>o</sub>	IOF <sub>c</sub>	h	k	l	IOF <sub>o</sub>	IOF <sub>c</sub>	h	k	l	IOF <sub>o</sub>	IOF <sub>c</sub>	h	k	l	IOF <sub>o</sub>	IOF <sub>c</sub>									
1	0	0	960	909	-2	1	1	530	575	-1	9	1	160	174	2	2	25	25	5	10	2	3	7	1	5	3	12	-30
			610	-630				76	64				191	-179			15	-12		11	11	43	40	-1	3	429	265	
			671	-619				162	162				162	162			15	-14		11	11	43	40	-1	3	297	-204	
			337	-362				107	88				110	-101			35	-26		11	11	270	249	-5	3	119	106	
			113	-103				239	-255				110	91			15	-8		11	11	312	312	-4	4	19	4	
			71	70				54	54				11	27			12	-37		3	3	724	206	-1	1	218	-197	
			151	-149				225	-233				12	-37			488	-515		3	3	532	531	-3	3	39	41	
			107	-91				56	54				10	51			212	219		1	1	67	52	-1	1	237	-231	
			67	50				98	84				110	-81			71	68		1	1	98	91	-1	1	121	-118	
			62	50				39	34				110	-83			110	84		1	1	43	43	-1	1	84	-77	
			11	17				680	-656				464	-179			297	304		1	1	141	-149	-1	1	344	-340	
			76	69				116	125				278	-291			193	-206		1	1	6	-42	10	1	17	-1	
			48	45				129	128				203	216			107	74		1	1	148	-173	-1	1	156	166	
			158	155				124	-91				158	141			143	-151		12	12	35	33	-1	1	168	-126	
			146	-162				148	116				199	-193			80	64		12	12	98	99	-1	1	150	132	
			98	-115				8	8				263	216			107	74		1	1	13	-46	-1	1	198	-156	
			43	-30				182	179				254	-267			80	-85		12	12	134	133	-1	1	150	112	
			50	-11				141	-125				110	120			13	-23		1	1	71	-53	-1	1	132	-128	
			67	52				195	205				184	-169			238	301		1	1	15	56	-1	1	25	25	
			305	309				136	-148				90	44			323	321		1	1	76	84	-1	1	191	-12	
			335	361				91	66				67	64			52	4		1	1	43	-44	-1	1	87	93	
			80	73				25	-23				12	-23			76	4		1	1	50	-36	-1	1	349	-340	
			80	-67				119	-83				12	-1			226	212		1	1	94	-54	-1	1	84	43	
			257	273				277	-278				11	-4			101	-126		1	1	13	-37	-1	1	17	-31	
			1086	1097				273	-278				11	-4			171	-175		1	1	13	-37	-1	1	361	353	
			337	-393				121	-135				10	32			300	-301		1	1	84	-82	-1	1	252	208	
			77	55				113	99				11	-29			62	-49		1	1	11	4	-1	1	152	146	
			307	310				215	-219				231	-217			75	37		1	1	57	54	-1	1	103	134	
			303	311				50	-28				84	68			10	-75		1	1	50	44	-1	1	19	19	
			127	124				67	36				121	114			80	83		1	1	62	-30	-1	1	91	-64	
			67	-149				56	-44				203	203			203	203		1	1	62	61	-1	1	50	-38	
			12	179				179	158				16	-96			187	-187		1	1	42	177	-1	1	58	58	
			281	297				35	-46				67	85			224	-179		1	1	91	90	-1	1	127	-85	
			119	122				59	84				91	-99			362	-293		1	1	43	-36	-1	1	182	-173	
			143	-138				204	218				91	-99			303	-266		1	1	43	-36	-1	1	127	-94	
			71	-71				138	166				80	-87			87	-60		1	1	10	-2	-1	1	17	-19	
			508	-540				176	-184				62	-58			77	-101		1	1	10	23	-1	1	19	-3	
			345	-368				127	-118				11	-124			101	-112		1	1	101	-112	-1	1	117	83	
			98	94				121	99				67	-62			80	77		1	1	124	-138	-1	1	56	31	
			236	303				193	-201				11	-41			71	74		1	1	98	-132	-1	1	50	36	
			188	194				224	233				11	-29			182	176		1	1	152	-148	-1	1	13	32	
			43	26				84	87				64	61			9	7		1	1	152	-148	-1	1	18	9	
			36	-67				107	108				10	-15			71	72		1	1	62	-68	-1	1	13	-6	
			71	71				67	-64				8	-74			110	339		1	1	9	-20	-1	1	246	-199	
			116	92				87	-89				6	-13			245	391		1	1	67	67	-1	1	91	-56	
			62	44				219	-255				182	179			245	391		1	1	9	35	-1	1	246	-201	
			80	-73				141	161				143	146			248	-252		1	1	87	-45	-1	1	218	170	
			116	92				295	-328				161	84			274	-272		1	1	17	17	-1	1	176	-168	
			56	-68				260	280				67	76			235	-191		1	1	514	-561	-1	1	215	190	
			204	199				109	-123				76	84			248	-252		1	1	514	-561	-1	1	18	-17	
			56	52				107	99				101	111			199	-173		1	1	685	-720	-1	1	145	125	
			87	87				124	-126				35	-70			67	59		1	1	197	155	-1	1	127	115	
			119	-103				67	12				79	35			67	59		1	1	279	279	-1	1	370	348	
			113	-112				141	142				35	-29			67	57		1	1	19	6	-1	1	307	304	
			10	-19				76	-75				80	-82			56	46		1	1	286	320	-1	1	136	96	
			290	-297				229	259				10	26			426	447		1	1	87	-50	-1	1	355	-351	
			261	-251				43	-50				59	135			159	135		1	1	269	279	-1	1	107	-85	
			29	-21				104	107				243	-215			248	-255		1	1	62	-36	-1	1	87	58	
			62	52				148	123				46	4			248	-255		1	1	184	-137	-1	1	134	-124	
			8	8				495	514				5	51			62	-41		1	1	184	-137	-1	1	370	348	
			43	37				8	22				156	162			228	-223		1	1	160	207	-1	1	94	-73	
			250	222				110	113				80	-92			12	-12		1	1	192	-170	-1	1	244	229	
			196	186				9	4				67	-64			110	98		1	1	188	234	-1	1	16	-18	
			10	-1				176	65				35	-42			11	6		1	1	113	120	-1	1	71	51	
			150	-133				209	-211				43	42			11	6		1	1	235	247	-1	1	139	110	
			124	-130				206	210				1215	-1228			177	168		1	1	168	170	-1	1	19	-24	
			62																									

Table 5 (cont.)

h	k	l	10F <sub>o</sub>	10F <sub>c</sub>	h	k	l	10F <sub>o</sub>	10F <sub>c</sub>	h	k	l	10F <sub>o</sub>	10F <sub>c</sub>	h	k	l	10F <sub>o</sub>	10F <sub>c</sub>	h	k	l	10F <sub>o</sub>	10F <sub>c</sub>
15	3	15	15	12	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
16	4	14	14	11	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
17	5	13	13	10	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
18	6	12	12	9	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
19	7	11	11	8	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
20	8	10	10	7	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
21	9	9	9	6	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
22	10	8	8	5	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
23	11	7	7	4	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
24	12	6	6	3	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
25	13	5	5	2	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
26	14	4	4	1	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
27	15	3	3	0	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
28	16	2	2	-1	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
29	17	1	1	-2	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
30	18	0	0	-3	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
31	19	-1	-1	-4	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
32	20	-2	-2	-5	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
33	21	-3	-3	-6	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
34	22	-4	-4	-7	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
35	23	-5	-5	-8	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
36	24	-6	-6	-9	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
37	25	-7	-7	-10	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
38	26	-8	-8	-11	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
39	27	-9	-9	-12	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
40	28	-10	-10	-13	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
41	29	-11	-11	-14	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
42	30	-12	-12	-15	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
43	31	-13	-13	-16	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
44	32	-14	-14	-17	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
45	33	-15	-15	-18	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
46	34	-16	-16	-19	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
47	35	-17	-17	-20	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
48	36	-18	-18	-21	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
49	37	-19	-19	-22	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
50	38	-20	-20	-23	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
51	39	-21	-21	-24	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
52	40	-22	-22	-25	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
53	41	-23	-23	-26	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
54	42	-24	-24	-27	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
55	43	-25	-25	-28	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
56	44	-26	-26	-29	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
57	45	-27	-27	-30	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
58	46	-28	-28	-31	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
59	47	-29	-29	-32	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
60	48	-30	-30	-33	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
61	49	-31	-31	-34	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
62	50	-32	-32	-35	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
63	51	-33	-33	-36	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
64	52	-34	-34	-37	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
65	53	-35	-35	-38	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
66	54	-36	-36	-39	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
67	55	-37	-37	-40	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
68	56	-38	-38	-41	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
69	57	-39	-39	-42	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
70	58	-40	-40	-43	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
71	59	-41	-41	-44	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
72	60	-42	-42	-45	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
73	61	-43	-43	-46	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
74	62	-44	-44	-47	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
75	63	-45	-45	-48	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
76	64	-46	-46	-49	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
77	65	-47	-47	-50	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
78	66	-48	-48	-51	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
79	67	-49	-49	-52	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
80	68	-50	-50	-53	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
81	69	-51	-51	-54	8	4	280	-270	4	5	152	171	-155	10	6	100	100	5	5	10	5	10	5	10
82	70	-52	-52	-55	8	4	280	-270	4	5	152	171	-155	10</										

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### Sodium Silicate Hydrates. III. The Crystal Structure of $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 6\text{H}_2\text{O}$ and of the Isostructural $\text{Na}_2\text{O} \cdot \text{GeO}_2 \cdot 6\text{H}_2\text{O}$ .

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A new sodium germanate hydrate, of oxide formula  $\text{Na}_2\text{O} \cdot \text{GeO}_2 \cdot 6\text{H}_2\text{O}$ , has been prepared and found to be isostructural with the silicate  $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 6\text{H}_2\text{O}$ . A complete structure analysis of both compounds has been carried out with three-dimensional data, with a view to extending the structural knowledge of sodium silicate hydrates, comparing interatomic bond distances and investigating the hydrogen bond systems. The silicon and germanium atoms are each surrounded tetrahedrally by four oxygen atoms. The tetrahedra being interconnected by fairly short hydrogen bonds. At least some of the hydrogen atoms seem to show a statistical distribution such that between two and three are associated with each silicon-oxygen or germanium-oxygen tetrahedron. The sodium atoms are in approximately octahedral coordination, the octahedra alternately sharing faces and corners to form sheets.

#### Introduction

A previous structural study of  $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 9\text{H}_2\text{O}$  (Jamieson & Dent Glasser, 1966*b*) has shown that this compound contains  $(\text{H}_2\text{SiO}_4)^{2-}$  groups. As a continuation of work in this field, the structure of the hexahydrate was next investigated, to find if the anions differed and to compare bond distances.

As silicon, sodium and oxygen do not differ greatly in scattering power, it was thought that if an isostructural germanate could be prepared the structure determination would be simplified, because of the presence of the comparatively heavy germanium atom. Prior to this study, only one sodium germanate hydrate was known –  $\text{Na}_2\text{O} \cdot \text{GeO}_2 \cdot 7\text{H}_2\text{O}$  (Pugh, 1926; Nowotny & Szekely, 1952) – for which there is no corresponding silicate. Weight-loss determinations on  $\text{Na}_2\text{O} \cdot \text{GeO}_2 \cdot 7\text{H}_2\text{O}$  (Schwarz & Heinrich, 1932) suggest the existence of a hexahydrate.

#### Experimental

The  $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 6\text{H}_2\text{O}$  crystals used in this study were prepared as described in part I (Jamieson & Dent Glasser, 1966*a*). The crystals did not have very regular geometrical forms; the one selected for intensity measurement was not greater than 0.3 mm in its longest dimension.

For preparation of the germanate hydrate, mixtures of sodium hydroxide and germanium dioxide were first fused in a platinum crucible at 1250°C.  $\text{Na}_2\text{O} : \text{GeO}_2$  molar ratios were (i) 1.36:1 (ii) 1.57:1 and (iii) 1.84:1. The quenched material was divided into portions, a different volume of water being added to each. Many of the preparations yielded crystals of  $\text{Na}_2\text{O} \cdot \text{GeO}_2 \cdot 7\text{H}_2\text{O}$  as identified by X-ray single-crystal and powder photographs and checked by chemical analysis. One solution [4.5 g of melt (ii) in 8 ml of water] which for the silicate series would be expected to yield  $\text{Na}_2\text{O} \cdot \text{SiO}_2 \cdot 9\text{H}_2\text{O}$  was seeded with crystals of this. In a short time, very large clear crystals had grown, but as soon as these were handled in any way, they turned cloudy,

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